

of the molecule leads to r.m.s. distances for the corresponding atoms of 0.120 and 0.094 Å for molecule (I) and (II), respectively. The molecular packing of the compound, Fig. 2, displays no intermolecular approaches with distances less than the sum of the van der Waals radii.

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References

- BERRY, R. S. (1960). *J. Chem. Phys.* **32**, 933–938.
 CROMER, D. & MANN, J. (1968). *Acta Cryst.* **A24**, 321–324.
 GILLESPIE, P., HOFFMANN, P., KLUSACEK, H., MARQUARDING, D., PFOHL, S., RAMIREZ, F., TSOLIS, E. A. & UGI, I. (1971). *Angew. Chem.* **83**, 691–721.
 HIRSHFELD, F. L. & RABINOVICH, D. (1973). *Acta Cryst.* **A29**, 510–513.
 HOFFMANN, R., HOWELL, J. M. & MUETTERTIES, E. L. (1972). *J. Amer. Chem. Soc.* **94**, 3047–3058.
 JOHNSON, C. K. (1965). *ORTEP*. Report ORNL-3794, Oak Ridge National Laboratory, Oak Ridge, Tennessee.
 KEPERT, D. L. (1973). *Inorg. Chem.* **12**, 1942–1944.
 SPRATLEY, R. D., HAMILTON, W. C. & LADELL, J. (1967). *J. Amer. Chem. Soc.* **89**, 2272–2278.
 STEWART, R. F., DAVIDSON, E. R. & SIMPSON, W. T. (1965). *J. Chem. Phys.* **42**, 3175–3187.
 WIEBER, M. & HOOS, W. R. (1968). *Tetrahedron Lett.* **51**, 5333–5334.
 WUNDERLICH, H. & MOOTZ, D. (1973). First European Crystallographic Meeting, Bordeaux.
 WUNDERLICH, H. & MOOTZ, D. (1974). *Acta Cryst.* **B30**, 935–939.
 WUNDERLICH, H., MOOTZ, D., SCHMUTZLER, R. & WIEBER, M. (1974). *Z. Naturforsch.* In the press.
 X-RAY (1972). Technical Report TR-192 of the Computer Science Center, Univ. of Maryland, June 1972.
 ZEMANN, J. (1963). *Z. anorg. allgem. Chem.* **324**, 241–249.

Acta Cryst. (1974). **B30**, 945

The Two-Dimensional Tunnel Structures of $K_3Sb_5O_{14}$ and $K_2Sb_4O_{11}$ *

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The structures of $K_3Sb_5O_{14}$ and $K_2Sb_4O_{11}$ have been solved by the single-crystal X-ray direct method and the heavy-atom method, respectively. The structure of $K_3Sb_5O_{14}$ is orthorhombic, with space group *Pbam* and cell parameters $a = 24.247$ (4), $b = 7.157$ (2), $c = 7.334$ (2) Å, $Z = 4$. The structure of $K_2Sb_4O_{11}$ is monoclinic, with space group *C2/m* and cell parameters $a = 19.473$ (4), $b = 7.542$ (1), $c = 7.198$ (1) Å, $\beta = 94.82$ (2)°, $Z = 4$. A full-matrix least-squares refinement gave $R = 0.072$ and $R = 0.067$, respectively. In both structures, oxygen atoms form an octahedron around each Sb atom and an irregular polyhedron around each K atom. By sharing corners and edges, the octahedra form a skeleton network having intersecting *b*-axis and *c*-axis tunnels. The K^+ ions, which have more than ten oxygen near neighbors, are located in these tunnels. Evidence for K^+ -ion transport within and between tunnels comes from ion exchange of the alkali ions in molten salts and anisotropic temperature factors that are anomalously large in the directions of the tunnels.

Introduction

Unlike Nb^{5+} and Ta^{5+} , the Sb^{5+} ion does not form structures having 180° Sb–O–Sb linkages (Goodenough & Kafalas, 1973). Thus $KSbO_3$ does not form the cubic perovskite structure. At atmospheric pressure it generally has the rhombohedral ilmenite structure. However, Spiegelberg (1940) reported synthesiz-

ing two cubic phases of $KSbO_3$ by annealing the ilmenite $KSbO_3$ for 3 weeks at 1000°C in a porcelain crucible. The structure of one of these, *Pn3*, was determined by him. The structure of the other, *Im3*, was elucidated by the present author (Hong, Kafalas & Goodenough, 1973). Both cubic structures contain Sb_2O_{10} edge-shared octahedra. These octahedral-site pairs share corners to form a three-dimensional skeleton structure containing empty tunnels of face-shared octahedra that run parallel to the cubic diagonals. In the *Pn3* structure, the K^+ ions are ordered on the

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octahedral sites of these tunnels, while in $Im\bar{3}$ the K^+ ions are distributed randomly on these octahedral sites. The $Im\bar{3}$ - $KSbO_3$ structure can be ion-exchanged (Hong *et al.*, 1973) to $MSbO_3$ ($M=Li, Na, Rb, Ag, Tl$) in molten salt.

The Sb^{5+} ion, preferring to form pairs of edge-shared oxygen octahedra, should produce other examples of three-dimensional networks containing large-mouth tunnels or voids. Such skeleton structures are of potential interest for fast-ion transport (Hong *et al.*, 1973). We have therefore synthesized two compounds, $K_3Sb_5O_{14}$ and $K_2Sb_4O_{11}$, to demonstrate other skeleton structures built by these Sb_2O_{10} pairs of edge-shared octahedra. The structure of $K_3Sb_5O_{14}$ was solved by the direct method in space group $Pbam$, which is different from that previously reported (Aurivillius, 1966) as $Pba2$. The $K_2Sb_4O_{11}$ compound was synthesized and its structure determined for the first time.

Experimental procedure

Single crystals of $K_3Sb_5O_{14}$ were obtained by first heating overnight at $800^\circ C$ in platinum crucibles K_2CO_3 and Sb_2O_3 in the molar ratio 3:5. The product was then ground and fired at $1100^\circ C$ for two days. A number of square-plate, single crystals were formed in this preparation. The thickness of a plate, which is the direction of the longest axis, was about one-third of its square edge.

The compound $K_2Sb_4O_{11}$ was discovered by a wet preparation. A saturated K_2CO_3 water solution was neutralized by $SbCl_5$. The white precipitate was evaporated slowly to dryness, then fired up to $600^\circ C$ overnight, ground and refired at $1100^\circ C$ for two days. The single crystals formed in this procedure have a crystal habit similar to that of $K_3Sb_5O_{14}$. The square edge was about 0.1 mm to 0.5 mm in both compounds.

For X-ray diffraction studies, a small crystal of each compound was mounted on a goniometer head. The $K_3Sb_5O_{14}$ crystal measured $0.05 \times 0.15 \times 0.15$ mm and was mounted about the c axis (0.15 mm), while the $K_2Sb_4O_{11}$ crystal measured $0.03 \times 0.10 \times 0.10$ mm and was mounted about the b axis (0.1 mm). Oscillation and Weissenberg photographs showed diffraction symmetry mmm for $K_3Sb_5O_{14}$, and the systematic absences were $0kl$, $k=2n+1$, and $h0l$, $h=2n+1$, which is consistent with space groups $Pba2$ and $Pbam$. For $K_2Sb_4O_{11}$, the diffraction symmetry is $2/m$ and the systematic absences were hkl , $h+k=2n+1$, which is consistent with space groups Cm , $C2$ and $C2/m$.

Three-dimensional intensity data to $2\theta \leq 50^\circ$ were collected – a total of 952 independent reflections for $K_3Sb_5O_{14}$ and 1003 independent reflections for $K_2Sb_4O_{11}$ – by the stationary-crystal, stationary-counter method using Zr-filtered $Mo K\alpha$ radiation and a 5° take-off angle. Each peak was counted for 10 s and the background (at 2° below the 2θ value of the peak) was also counted for 10 s. Lorentz, polarization and φ -angle absorption corrections were applied. The

intensity variation on the φ angle was within 15% for $K_3Sb_5O_{14}$ and 20% for $K_2Sb_4O_{11}$. The equation

$$\sigma(F) = \frac{1}{2} \left[K \frac{1 + I_B/I_P}{1 - I_B/I_P} \right]^{1/2}$$

was used to estimate the variance for the structure factors, where I_B is the background count, I_P is the peak count, and K is the product of the Lorentz, polarization and absorption corrections.

Structure determinations

$K_3Sb_5O_{14}$

The direct method for determination of phases was used in the solution of this structure. First, a Wilson plot was constructed. This plot provided an approximate scale factor and also indicated a centric distribution of intensities so that the space group $Pbam$ was chosen. The sequence of computer programs *FAME-MAGIC-LINK-SYMP* (R. Dewar & A. Stone, Univ. of Chicago) was used to generate 250 phases from seven symbol-assigned reflections. Two E maps were generated using different sign combinations for the symbolic assignments, and the correct map was ascertained by checking interactions between the largest peaks against the Patterson map. The correct E map displayed a uniform distribution of peak heights, and the heights were in the appropriate ratio of the atomic numbers of the atoms. Two independent K and three independent Sb positions were clearly visible; the oxygen atoms could not be unambiguously assigned as there were a number of spurious peaks with heights equivalent to those of the oxygen atoms. A structure-factor calculation based on the K and Sb positions gave a value of 0.18 for the discrepancy index $R=$

Table 1. Final atomic parameters and isotropic temperature factors in $K_3Sb_5O_{14}$

Space group: $Pbam$, $Z=4$.
Unit-cell dimensions:
 $a=24.247$ (4), $b=7.157$ (2), $c=7.334$ (2) Å.
Standard deviations are given in parentheses.

	x	y	z	β (Å ²)
K(1)	0.0152 (6)	0.219 (2)	$\frac{1}{2}$	4.7 (3)
K(2)	0.2934 (6)	0.111 (2)	0	5.6 (3)
K(3)	0.3839 (6)	0.102 (2)	$\frac{1}{2}$	4.5 (3)
Sb(1)	0.0570 (1)	0.4169 (5)	0	0.51 (5)
Sb(2)	0.4379 (1)	0.4045 (5)	0	0.60 (5)
Sb(3)	0.2564 (1)	0.3249 (4)	$\frac{1}{2}$	0.51 (5)
Sb(4)	0.14506 (9)	0.1086 (3)	0.2646 (3)	0.51 (4)
O(1)	0	$\frac{1}{2}$	0.159 (4)	2.7 (7)
O(2)	0.017 (1)	0.156 (4)	0	1.6 (6)
O(3)	0.117 (1)	0.175 (5)	$\frac{1}{2}$	2.5 (7)
O(4)	0.136 (1)	0.024 (5)	0	2.8 (7)
O(5)	0.279 (1)	0.066 (4)	$\frac{1}{2}$	1.5 (6)
O(6)	0.410 (1)	0.154 (5)	0	2.9 (7)
O(7)	0.1046 (9)	0.336 (3)	0.188 (3)	2.0 (4)
O(8)	0.2096 (8)	0.254 (3)	0.284 (2)	1.4 (3)
O(9)	0.3150 (8)	0.379 (3)	0.316 (2)	1.6 (4)
O(10)	0.4249 (8)	0.460 (3)	0.251 (3)	2.2 (4)

$\sum|F_o - F_c|/\sum|F_o|$. A Fourier map immediately revealed the positions of eight independent oxygen atoms. The atom parameters, scale and isotropic temperature fac-

Table 3. Observed and calculated structure factors for $K_3Sb_5O_{14}$

Table with columns: h, k, Fobs, Fcalc, h, k, Fobs, Fcalc, h, k, Fobs, Fcalc, h, k, Fobs, Fcalc, h, k, Fobs, Fcalc. The table contains multiple rows of numerical data representing structure factors.

Table 2. Distances (Å) and angles (°) in $K_3Sb_5O_{14}$

Octahedron around Sb(1)			
Sb(1)-O(1)	2 × 1.90 (2)	O(1)-O(1)	2.33 (6)
Sb(1)-O(7)	2 × 1.89 (2)	O(1)-O(2)	2.76 (3)
Sb(1)-O(2)	2.10 (3)	O(1)-O(6)	2.71 (3)
Sb(1)-O(6)	1.88 (3)	O(1)-O(7)	2.80 (2)
Sb(1)-Sb(2)	3.009 (5)	O(2)-O(6)	3.89 (4)
Sb(1)-Sb(2)	3.492 (5)	O(2)-O(7)	2.84 (3)
Sb(1)-Sb(4)	3.632 (3)	O(6)-O(7)	2.68 (4)
Octahedron around Sb(2)			
Sb(2)-O(10)	2 × 1.91 (2)	O(2)-O(2)	2.38 (6)
Sb(2)-O(2)	1.97 (2)	O(2)-O(4)	3.93 (4)
Sb(2)-O(2)	2.11 (3)	O(2)-O(6)	2.93 (4)
Sb(2)-O(4)	1.99 (3)	O(2)-O(10)	2.71 (3)
Sb(2)-O(6)	1.92 (3)	O(4)-O(6)	2.87 (5)
Sb(2)-Sb(4)	3.154 (3)	O(4)-O(10)	2.40 (3)
Sb(2)-Sb(2)	3.307 (5)	O(6)-O(10)	2.83 (3)
Octahedron around Sb(3)			
Sb(3)-O(8)	2 × 2.01 (2)	O(5)-O(5)	3.84 (2)
Sb(3)-O(9)	2 × 2.00 (2)	O(5)-O(8)	2.68 (3)
Sb(3)-O(5)	1.93 (3)	O(5)-O(8)	2.75 (3)
Sb(3)-O(5)	1.93 (3)	O(5)-O(9)	2.97 (3)
Sb(3)-Sb(3)	3.592 (1)	O(5)-O(9)	2.76 (3)
Sb(3)-Sb(4)	3.558 (3)	O(8)-O(9)	2.72 (3)
Sb(3)-Sb(4)	3.579 (3)	O(8)-O(9)	2.76 (3)
Octahedron around Sb(4)			
Sb(4)-O(3)	1.92 (1)	O(3)-O(7)	2.58 (3)
Sb(4)-O(4)	2.04 (1)	O(3)-O(8)	2.80 (3)
Sb(4)-O(7)	1.98 (2)	O(3)-O(9)	3.00 (3)
Sb(4)-O(8)	1.88 (2)	O(3)-O(10)	2.60 (3)
Sb(4)-O(9)	1.94 (2)	O(4)-O(7)	2.73 (4)
Sb(4)-O(10)	2.00 (2)	O(4)-O(8)	3.19 (3)
Sb(4)-Sb(2)	3.154 (3)	O(4)-O(9)	2.80 (2)
Sb(4)-Sb(1)	3.632 (3)	O(4)-O(10)	2.40 (3)
Sb(4)-Sb(3)	3.558 (3)	O(7)-O(10)	2.82 (3)
Sb(4)-Sb(3)	3.579 (3)	O(9)-O(10)	2.77 (3)
Sb-O-Sb angles			
Sb(1)-O(2)-Sb(2)	121 (1)	Sb(1)-O(1)-Sb(1)	104 (1)
Sb(2)-O(2)-Sb(2)	108 (1)	Sb(4)-O(3)-Sb(4)	129 (2)
Sb(2)-O(2)-Sb(1)	129 (1)	Sb(3)-O(5)-Sb(3)	137 (1)
Sb(2)-O(4)-Sb(4)	103.0 (7)	Sb(1)-O(6)-Sb(2)	134 (1)
Sb(4)-O(4)-Sb(6)	143 (2)	Sb(1)-O(7)-Sb(4)	139 (1)
Sb(6)-O(4)-Sb(2)	103.0 (7)	Sb(3)-O(8)-Sb(4)	131.7 (9)
		Sb(3)-O(9)-Sb(4)	130 (1)
		Sb(2)-O(10)-Sb(4)	107 (1)
Polyhedron around K(1)			
K(1)-O(1)	2 × 3.23 (2)	Polyhedron around K(2)	
K(1)-O(2)	2 × 3.695 (4)	K(2)-O(4)	1 × 3.42 (4)
K(1)-O(3)	1 × 2.49 (3)	K(2)-O(5)	2 × 3.698 (4)
K(1)-O(7)	2 × 3.26 (2)	K(2)-O(6)	1 × 2.84 (3)
K(1)-O(10)	2 × 2.98 (2)	K(2)-O(7)	2 × 3.44 (2)
K(1)-O(10)	2 × 3.13 (2)	K(2)-O(8)	2 × 3.08 (2)
		K(2)-O(8)	2 × 3.29 (2)
		K(2)-O(9)	2 × 3.06 (2)
Polyhedron around K(3)			
K(3)-O(1)	2 × 3.83 (2)		
K(3)-O(3)	1 × 3.06 (4)		
K(3)-O(5)	1 × 2.56 (3)		
K(3)-O(6)	2 × 3.740 (6)		
K(3)-O(7)	2 × 2.99 (2)		
K(3)-O(8)	2 × 3.72 (2)		
K(3)-O(9)	2 × 2.93 (2)		
K(3)-O(10)	2 × 3.30 (2)		

tors were then refined with a full-matrix, least-squares program to a final R value of 0.072 for all reflections and a weighted $R_w = 0.067$, where

$$R_w = \left[\sum w(F_o - F_c)^2 / \sum wF_o^2 \right]^{1/2}, \quad w = 1/\sigma^2.$$

After this structure was solved, the author learned that Aurivillius (1966) had reported $K_3Sb_5O_{14}$ as having space group $Pba2$. Based on this space group, a least-squares refinement on our data gave $R = 0.075$ and $R_w = 0.069$ with negative temperature factors for those atomic positions having z coordination other than 0 or $1/2$. This indicates that the c axis contains mirror planes at $z=0$ and $z=1/2$. Thus three pieces of evidence indicate that $Pbam$ is the right space group: (1) a Wilson plot showed a center of symmetry, (2) with

more variable parameters $Pba2$ gave the higher R index, (3) negative temperature factors appeared on those atomic positions that should have a mirror-symmetry operation.

The final atomic parameters and isotropic temperature factors are listed in Table 1. The bond distances and angles are listed in Table 2. Table 3 contains the structure factors.

$K_2Sb_4O_{11}$

The composition of the $K_2Sb_4O_{11}$ crystals was initially unknown. Nevertheless, because Sb is considerably heavier than either K or O, it was possible to solve the structure by using the heavy-atom method of structure analysis. A three-dimensional Patterson map was generated to distinguish between the three possible space groups $C2$, Cm and $C2/m$. If Cm is correct,

Table 4. Final atomic parameters and anisotropic temperature factors in $K_2Sb_4O_{11}$

	x	y	z	β_{11}	β_{22}	β_{33}	β_{12}	β_{13}	β_{23}
Space group: $C2/m$, $Z=4$.									
Unit-cell dimensions: $a=19.473$ (4), $b=7.542$ (1), $c=7.198$ (1) Å, $\beta=94.82$ (2)°.									
The temperature factor is $\exp[-(\beta_{11}h^2 + \beta_{22}k^2 + \beta_{33}l^2 + 2\beta_{12}hk + 2\beta_{13}hl + 2\beta_{23}kl)]$.									
K(1)	0.2383 (9)	0	0.005 (2)	0.0022 (3)	0.017 (2)	0.013 (2)	0	0.0010 (7)	0
K(2)	0.0103 (6)	0	0.261 (1)	0.0022 (3)	0.019 (3)	0.018 (2)	0	0.0009 (7)	0
Sb(1)	0.4246 (1)	0	0.3824 (3)	0.00014 (6)	0.0001 (4)	0.0000 (4)	0	0.0000 (1)	0
Sb(2)	0.4281 (1)	0	0.9029 (3)	0.00016 (6)	0.0001 (4)	0.0001 (4)	0	0.0000 (1)	0
Sb(3)	0.17445 (8)	0.2418 (2)	0.4349 (2)	0.00004 (4)	0.0006 (3)	0.0003 (2)	0.0000 (1)	0.00008 (7)	0.0000 (2)
O(1)	0	0.335 (3)	0	0.0006 (6)	0.002 (4)	0.011 (5)	0	0.000 (1)	0
O(2)	0.160 (1)	0	0.348 (3)	0.0009 (9)	0.011 (6)	0.009 (5)	0	0.000 (1)	0
O(3)	0.382 (1)	0	0.131 (4)	0.0013 (9)	0.030 (9)	0.007 (5)	0	0.000 (1)	0
O(4)	0.335 (1)	0	0.483 (3)	0.0007 (8)	0.011 (6)	0.007 (5)	0	0.000 (1)	0
O(5)	0.522 (1)	0	0.353 (3)	0.003 (1)	0.012 (6)	0.003 (5)	0	0.001 (1)	0
O(6)	0.412 (1)	0.257 (3)	0.433 (2)	0.0028 (7)	0.005 (3)	0.011 (4)	0.000 (1)	0.000 (1)	0.002 (3)
O(7)	0.2654 (9)	0.296 (2)	0.342 (2)	0.0017 (6)	0.009 (4)	0.006 (3)	0.000 (1)	0.000 (1)	0.001 (3)
O(8)	0.1296 (9)	0.309 (2)	0.189 (2)	0.0013 (5)	0.005 (3)	0.011 (3)	0.000 (1)	0.001 (1)	0.003 (3)

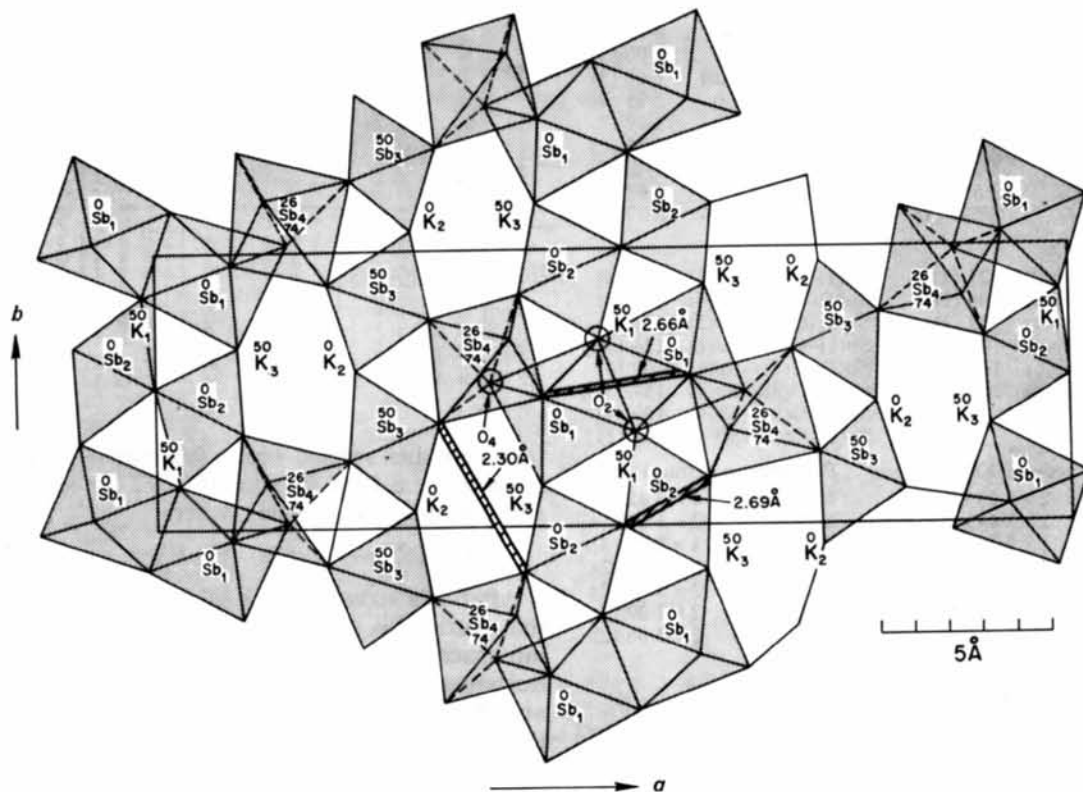


Fig. 1. An ab projection of the arrangement of SbO_6 octahedra in $K_3Sb_5O_{14}$, showing the tunnels that run parallel to the b axis and the c axis. The shaded bars represent the narrowest point along or between tunnels.

then a large number of interactions should be observed on the section [0,y,0]; and this was not the case: (0, 1/2, 0) is the only major peak. If C2 is the right space group, there is no mirror symmetry to produce the (0, 1/2, 0) peak, so C2/m was chosen.

On the Patterson map, all major peaks appeared at y equal to 0, 1/4 and 1/2, which indicated that the heavy atom Sb would be located at special positions. Based on the prior knowledge that Sb readily forms pairs of edge-shared octahedra, it was possible to build a model for testing. Three independent Sb positions were located to give R=0.24. A Fourier map based on this model revealed two independent K positions and eight O atoms. The atomic positions, scale and anisotropic temperature factors were then refined with a full-matrix, least-squares program to a final R=0.067 and R_w=0.080 for all reflections. A three-dimensional electron-density difference map calculated with the par-

Table 6. Observed and calculated structure factors for K2Sb4O11

Table with multiple columns containing numerical data for observed and calculated structure factors for K2Sb4O11. The table is organized into several sections based on h, k, l indices.

Table 5. Distances (Å) and angles (°) in the structure of K2Sb4O11

Table listing distances and angles in the structure of K2Sb4O11. It is divided into sections for Octahedron around Sb(1), Octahedron around Sb(2), Octahedron around Sb(3), and Sb-O-Sb angles.

ameters in Table 4 showed no physically significant peaks. It was then evident that the compound had the chemical formula K2Sb4O11 with four molecules per unit cell. The compound K2Sb4O11 can also be synthesized by heating a mixture of K2CO3 and 2Sb2O3 at 1100°C for two days. The bond distances and angles are listed in Table 5. Table 6 contains the observed and calculated structure factors.

Description of the structures

K3Sb5O14

The K3Sb5O14 structure is as described by Aurivill-

Table listing distances and angles for Polyhedron around K(1).

Table listing distances and angles for Polyhedron around K(2).

lius (1966) except for a mirror symmetry perpendicular to the c axis. It contains four $K_3Sb_5O_{14}$ molecules with Sb(1) and Sb(2) atoms at $z=0$. As shown in Fig. 1, Sb(1)–Sb(1) pairs of octahedra have a common edge in the ab plane centered at a body corner or the center of an ab face. Along the b axis, the Sb(1)–Sb(1) pairs are joined by corner-sharing with Sb(2)–Sb(2) pairs having their common edge parallel to the c axis. Chains of corner-shared Sb(3) octahedra centered at $z=\frac{1}{2}$ also run parallel to the b axis. The Sb(1)–Sb(2) ribbons and Sb(3) chains are connected only by Sb(4) octahedra centered at $z=26, 74$ and sharing a common edge with an Sb(1) octahedron, common corners with an Sb(2) and two Sb(3) octahedra.

The oxygen atoms O(2) and O(4) are each common to three octahedra (see circles in Fig. 1). The O(4) atom is common to two corner-shared Sb(4) octahedra, one above the other along z , as well as to an Sb(1) octahedron. All other oxygen atoms are common to only two octahedra. This sharing of the oxygen atoms is compatible with an O/Sb ratio of 2.8.

The $(Sb_5O_{14})^{3-}$ skeleton has tunnels parallel to the c axis that contain, zigzag fashion, the K^+ -ion positions K(2) and K(3). These c -axis tunnels are intersected by b -axis tunnels containing positions K(1). The

position K(3) can be considered common to a c -axis and a b -axis tunnel.

$K_2Sb_4O_{11}$

The $(Sb_4O_{11})^{2-}$ skeleton of the $K_2Sb_4O_{11}$ structure is entirely built from pairs of edge-shared octahedra. The Sb(1)–Sb(1) and Sb(3)–Sb(3) pairs have their common edge in the ac plane whereas the Sb(2)–Sb(2) pairs have their common edge parallel to the b axis, as can be seen from Figs. 2 and 3. The Sb(1)–Sb(1) and Sb(2)–Sb(2) pairs share common corners to form ribbons at $y=0, \frac{1}{2}$ running parallel to the c axis at $x=\frac{1}{2}$ and 0, respectively. These ribbons are bridged by Sb(3)–Sb(3) pairs, which form corner-shared columns parallel to the b axis. The Sb(3)–Sb(3) pairs share common octahedral-site edges with Sb(1)–Sb(1) pairs, common corners with Sb(2)–Sb(2) pairs. This skeleton structure produces tunnels parallel to both the b and c axes, and the K^+ ions are located in these tunnels. The irregular polyhedra around the K(1) and K(2) positions contain more than ten oxygen atoms, and the K–O distances are all longer than 2.76 Å, the sum of the ionic radii. From Table 5, the shortest K–O distance is K(1)–O(3)=2.86 Å.

From Table 4, the temperature factors are anoma-

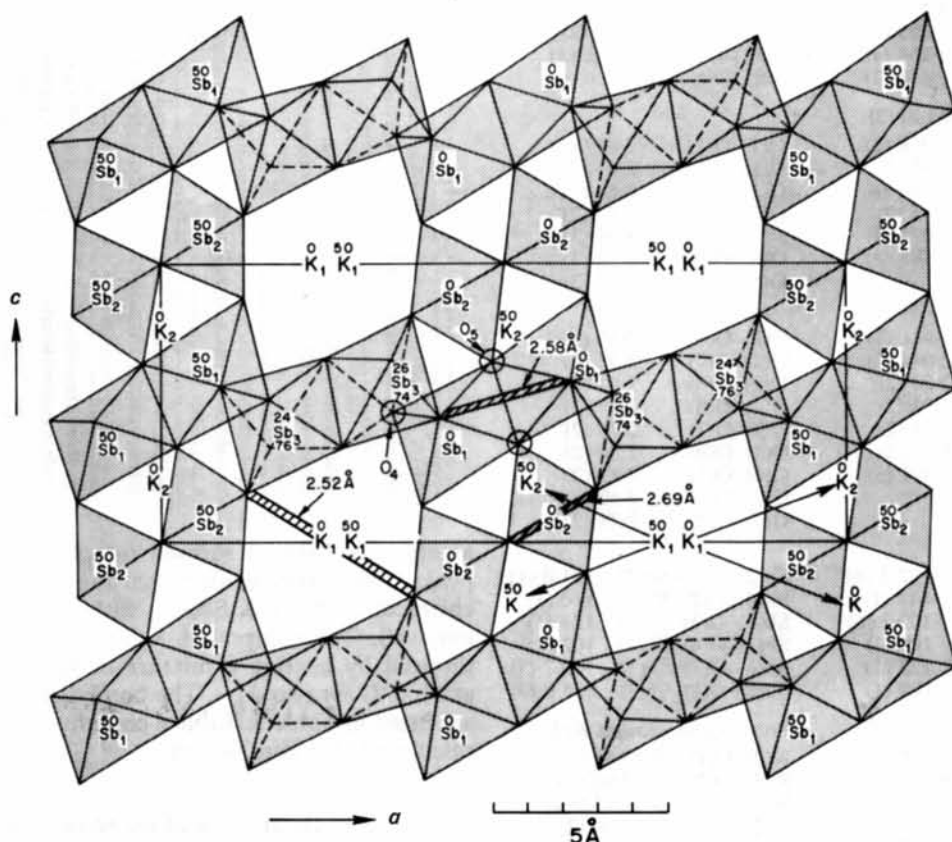


Fig. 2. An ac projection of the arrangement of SbO_6 octahedra in $K_2Sb_4O_{11}$, showing the tunnels formed along the b axis. The shaded bars are the narrowest points along or between tunnels.

lously larger for K(1) and K(2) than for the Sb atoms. Moreover, the thermal motions of K(1) and K(2) along the b axis, which are about the same as those along the c axis, are eight times bigger than those along the a axis. This anomalous thermal motion along the b and c axes indicates that a K^+ ion can move within and between tunnels.

The O(4) and O(5) anions (see circles in Fig. 2) are common to three octahedra: two Sb(3) and an Sb(1); two Sb(1) and one Sb(2), respectively. All the other oxygen atoms are common to only two octahedra. This type of sharing is consistent with an O/Sb ratio of $\frac{1}{4}$.

Discussion

The $K_3Sb_5O_{14}$ and $K_2Sb_4O_{11}$ structures have two-dimensional tunnels. In both structures, the K^+ ions are located in intersecting b -axis and c -axis tunnels. Although the K positions are fully occupied, there are a number of interstitial voids existing in the skeleton, so that K^+ ions may be transported in and between tunnels. These interstitial positions, large K-O distances,

and anomalous temperature factors are consistent with the fact that both compounds can be ion-exchanged by Na^+ , Ag^+ , Rb^+ and Tl^+ in molten salts. The skeletons are all intact after ion-exchange, except that $K_3Sb_5O_{14}$ transforms to $Na_2Sb_4O_{11}$ in molten $NaNO_3$. This transformation is apparently due to the fact that the Na^+ ion is too small to support the $K_3Sb_5O_{14}$ skeleton, especially as the $K_2Sb_4O_{11}$ skeleton is intrinsically more stable because all its octahedra are paired by edge-sharing.

The narrowest point along the tunnels, designated herein as 'bottle-necks', can be used as a criterion for fast-ion transport. In $K_3Sb_5O_{14}$, the bottle-neck along the b tunnels, the c tunnels and from a b tunnel to a c tunnel are rectangular planes having shortest K-O distances of 2.66, 2.30, and 2.69 Å, respectively. (In Fig. 1, the bottle-necks are indicated by the heavy lines.) From these distances we can expect the K^+ ions to move within the b tunnel or from a b tunnel to a c tunnel, but they will transport along the c tunnels with a higher activation energy. In $K_2Sb_4O_{11}$, the bottle-neck along the b tunnels and the c tunnels and from a

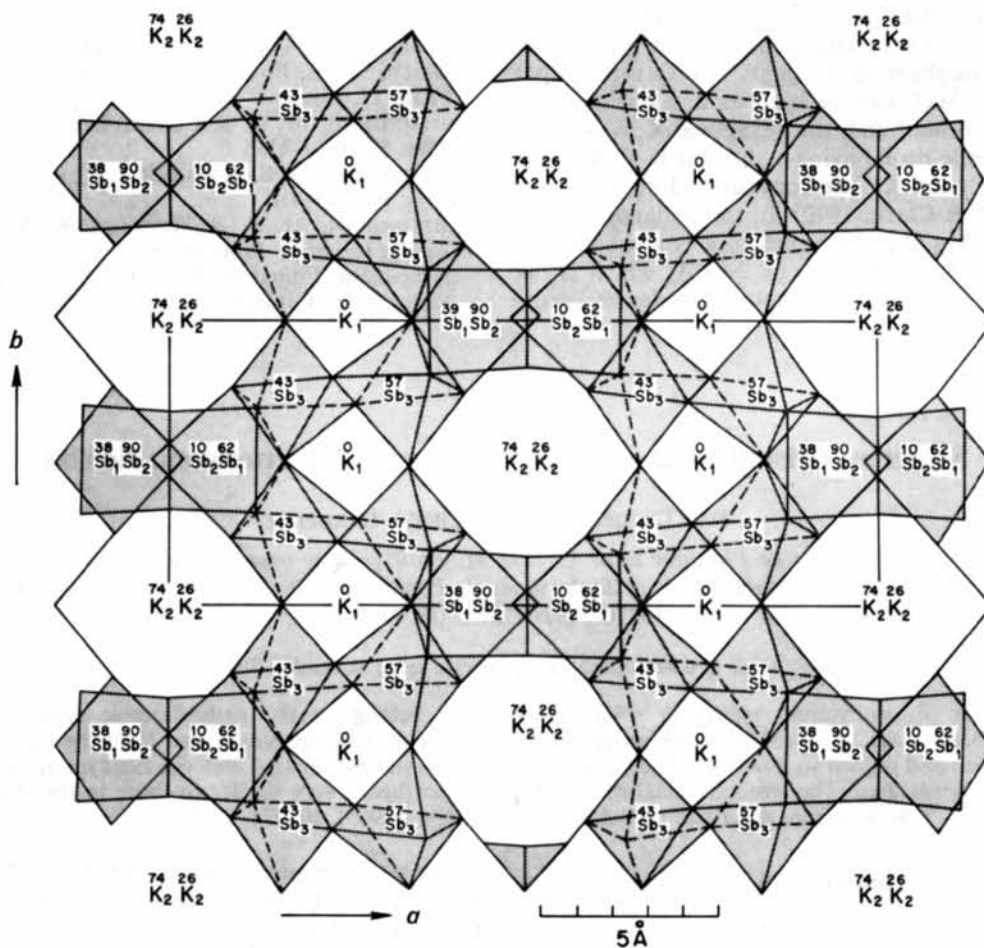


Fig. 3. An ab projection of the arrangement of SbO_6 octahedra in $K_2Sb_4O_{11}$, showing the tunnels that run parallel to the c axis.

b tunnel to a c tunnel are rectangular planes having shortest K–O distances of 2.58, 2.52, and 2.69 Å, respectively, as shown in Fig. 2. Here we would expect the K^+ ions to transport readily along both the b and c axes.

The same criterion can be applied to all tunnel structures. For instance, in the cubic perovskite structure, the bottle-neck is a square along the tunnels parallel to the principal axes. The alkali–oxygen distance in this square is half of the axis, or about 1.9 Å, which is just big enough for H–O bonding or Li–O bonding. This may explain why Vannice, Boudart & Fripiat (1970) reported that the protons in high-surface-area H_xWO_3 were mobile at liquid-nitrogen temperature and why Whittingham & Huggins (1972) found that Li_xWO_3 can be used as a lithium-ion reversible electrode, whereas Smith & Danielson (1954) found an extremely low diffusion coefficient for cubic Na_xWO_3 even at elevated temperatures. Tetragonal K_xWO_3 (Magnéli, 1949) has two kinds of tunnels running along the c axis; one has a square bottle-neck similar in size to the cubic perovskite; another is a pentagon bottle-neck with a K–O distance of 2.17 Å. It is obvious that K^+ ions cannot move easily along these tunnels. Even a Na^+ ion is too big to pass through the bottle-neck without some activation energy. Nevertheless, Whittingham & Huggins (1971) report that tetragonal Na_xWO_3 can be used as a reversible electrode. On the other hand, the hexagonal K_xWO_3 , with a six-sided bottle-neck having a shortest K–O distance of 2.87 Å, is a good K^+ -ion conductor along the c axis (Whittingham & Clark, 1970). In the hollandite structure, $Ba_xTi_{8-x}Mg_xO_{16}$, the bottle-neck is a square having a shortest Ba–O distance of 2.60 Å, which is big enough for a Ba^{2+} ion to transport (Dryden & Wads-

ley, 1953). The $Im3-NaSbO_3$ structure has a three-sided bottle-neck along the diagonal tunnel with a shortest Na–O distance of 2.40 Å, and the Na^+ ion is mobile (Hong *et al.*, 1973).

Whereas one-dimensional tunnels are probably not practical for polycrystalline membranes, because of interruption at grain boundaries, structures having two-dimensional and three-dimensional tunnel networks could be suitable.

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References

- AURIVILLIUS, B. (1966). *Ark. Kem.* **25**, 505–514.
 DRYDEN, J. S. & WADSLEY, A. D. (1953). *Trans. Faraday Soc.* **54**, 1574–1580.
 GOODENOUGH, J. B. & KAFALAS, J. A. (1973). *J. Solid State Chem.* **6**, 493–501.
 HONG, H. Y.-P., KAFALAS, J. A. & GOODENOUGH, J. B. (1973). *J. Solid State Chem.* In the press.
 MAGNÉLI, A. (1949). *Ark. Kem.* **1**, 213–221.
 SMITH, J. F. & DANIELSON, G. E. (1954). *J. Chem. Phys.* **22**, 266.
 SPIEGELBERG, P. (1940). *Ark. Kem.* **14A**, 1–12.
 VANNICE, M. A., BOUDART, M. & FRIPIAT, J. J. (1970). *J. Catal.* **17**, 359.
 WHITTINGHAM, M. S. & CLARK, L. D. (1970). *J. Chem. Phys.* **53**, 4114–4115.
 WHITTINGHAM, M. S. & HUGGINS, R. A. (1971). *J. Chem. Phys.* **54**, 414–416.
 WHITTINGHAM, M. S. & HUGGINS, R. A. (1972). *Solid State Chemistry, Proceedings of 5th Materials Research Symposium*. Edited by R. S. ROTH & S. J. SCHNEIDER. NBS Special Publication 364, pp. 139–154.

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Structure Cristalline du Sulfure de Néodyme et d'Ytterbium, $NdYbS_3$

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Crystals of neodymium ytterbium sulphide, $NdYbS_3$, belong to the orthorhombic system, $a=12.545$, $b=9.438$, $c=3.853$ Å, space group $B22_12$, $Z=4$. The structure was solved from the Patterson function and refined to $R=0.047$. The ytterbium atoms are six-coordinated and the neodymium atoms eight-coordinated. The structure exhibits coordination polyhedra very similar to those in the yttrium sulphide, Y_3S_7 , structure described by Adolphe [*Ann. Chim.* (1965). **10**, 271–275].

Introduction

L'étude systématique des sulfures doubles de terres rares a conduit Khodadad & Vo Van Tien (1969) à isoler une phase nouvelle à partir d'un mélange en

quantité équimoléculaire de Yb_2S_3 et de Nd_2S_3 . Ce mélange est chauffé en tube scellé à 1000°C pendant deux jours. Des cristaux suffisamment gros s'obtiennent en laissant pendant un mois, à 850°C, de la poudre cristalline dans du bromure de potassium fondu.